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3 **RELATION BETWEEN FIBRE DISTRIBUTION AND POST-CRACKING BEHAVIOUR IN**
4 **STEEL FIBRE REINFORCED SELF-COMPACTING CONCRETE PANELS**
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12 **ABSTRACT**

13 In this research, the influence of the fibre distribution and orientation on the post-cracking behaviour of steel fibre
14 reinforced self-compacting concrete (SFRSCC) panels was studied. To perform this evaluation, SFRSCC panels
15 were cast from their centre point. For each SFRSCC panel, cylindrical specimens were extracted and notched either
16 parallel or perpendicular to the concrete flow direction, in order to evaluate the influence of fibre dispersion and
17 orientation on the tensile performance. The post-cracking behaviour was assessed by both splitting tensile tests and
18 uniaxial tensile tests. To assess the fibre density and orientation through the panels, an image analysis technique was
19 employed across cut planes on each tested specimen. It is found that the splitting tensile test overestimates the post-
20 cracking parameters. Specimens with notched plane parallel to the concrete flow direction show considerable higher
21 post-cracking strength than specimens with notched plane perpendicular to the flow direction.
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23 Keywords: Fibre reinforcement; Tensile properties; Rheology; Dispersion; Image analysis
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28 **1. INTRODUCTION**

29 The addition of fibres to a cementitious matrix contributes mainly to the energy absorption capacity and crack
30 control of structural elements, as well as to the enhancement of the load bearing capacity, particularly, in structural
31 configurations with high support redundancy [1-2]. The fibre reinforcement mechanisms are mainly effective after
32 concrete cracking initiation and, mostly, improve the post-cracking behaviour, due to the stress transfer provided by
33 fibres bridging cracked sections. Crack opening in steel fibre reinforced concrete (SFRC) is counteracted by the
34 bond stresses that develop at the fibres / matrix interface during the fibre pull-out. On the other hand, one of the
35 most important properties of SFRC is its ability to transfer stresses across a cracked section rather uniformly, which
36 nonetheless is dependent on the fibre reinforcement effectiveness, i.e. fibre properties (their strength, bond, and
37 stiffness), and fibre orientation and distribution [3]. The stress transfer capability of fibres enhances mainly the
38 composite's toughness, which is a parameter related to the energy absorption during monotonic or cyclic loading
39 [4].

40 The dispersion and orientation of fibres in the hardened-state results from a series of stages that SFRC passes from
41 mixing to hardening state, namely [5]: fresh-state properties after mixing; casting conditions into the formwork;
42 flowability characteristics; vibration and wall-effect introduced by the formwork. Among these factors, wall-effects
43 introduced by the moulds, and the properties of SFRC in the fresh state, especially its flowability, are the most
44 important ones [5-7]. Having in mind that mechanical properties are significantly related to the fibre orientation and
45 dispersion, which are affected by concrete's flow in the fresh state, it is important to control both those parameters
46 (flowability and wall-effect) [8-10].

47 Application of steel fibres enhances the mechanical properties of concrete, but since all fibres cannot be aligned in
48 the direction of the applied stress, the effectiveness of the fibres is dependent of the loading conditions, mainly on
49 the directions of the principal tensile stresses. Moreover, it is shown that the fibre distribution's scatter in large scale
50 elements may result in a significant inconsistency of the mechanical behaviour along the structural element.
51 Therefore, it is feasible to expect an anisometric material behaviour for this kind of composite. In addition, the fibre
52 efficiency depends on the orientation of the fibres towards the active crack plane. Some authors agree that in steel
53 fibre reinforced self-compacting concrete (SFRSCC) the variability in the post-cracking parameters observed in
54 bending tests, and also in uniaxial direct tensile tests, can be justified by the dispersion and alignment of the fibres
55 [11-12]. Therefore, a significant research effort has been done to achieve better mechanical performances for SFRC

56 by conditioning the distribution and orientation of the fibres [11, 13-15]. However, these effects should be
57 considered for structural design, especially when fibre distribution and orientation affect significantly the
58 mechanical properties of SFRC.

59 The main objective of this study is to connect experimentally the influence of the distribution / orientation of fibres,
60 which are affected by flowability of concrete, to the post-cracking behaviour of SFRSCC developed and applied on
61 laminar structures. To perform this evaluation SFRSCC panels were casted from their centre point. For each
62 SFRSCC panel, cylindrical specimens were extracted and notched either parallel or perpendicular to the concrete
63 flow direction to evaluate the effects of fibre dispersion and alignment on the tensile performance. The post-cracking
64 behaviour was assessed by both splitting tensile tests and also uniaxial tensile tests. To characterize fibre density and
65 orientation throughout the panels, an image analysis technique was employed across the cut plane of each tested
66 specimen.

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69 **2. EXPERIMENTAL RESEARCH**

70 **2.1 Concrete mixture**

71 The constituent materials used in the composition of the SFRSCC were: Portland cement CEM 42.5 R (C), water
72 (W), superplasticizer Sika® 3005 (SP), limestone filler, crushed granite aggregate, fine and coarse sand, and hooked-
73 end steel fibres (length, l_f , of 33 mm; diameter, d_f , of 0.55 mm; aspect ratio, l_f/d_f , of 60 and a yield stress of 1100
74 MPa). The adopted mix proportions are shown in Table 1, where W/C is the water/cement ratio. To evaluate the
75 properties of SFRSCC in the fresh state, the inverted Abrams cone slump test was performed according to EFNARC
76 recommendations [16]. An average spread of 670 mm was achieved without sign of segregation of the constituents.
77 The compressive strength and Young's modulus were determined using cylinders of 150 mm diameter and 300 mm
78 height after 28 days of moist curing in a climate chamber (3 cylinders for each test). The average compressive
79 strength (f_{cm}) and the average value of the Young's modulus (E_{cm}) were 47.77 MPa (7.45 %) and 34.15 GPa (0.21
80 %), respectively, where the values in parentheses represent the coefficient of variation.

81

82 2.2 Specimens

83 According to [17], casting a slab from its centre assures better mechanical behaviour compared to the other casting
84 methods. Therefore this direction of casting was selected for the production of two SFRSCC panels. The dimensions
85 of the panels are $1600 \times 1000 \text{ mm}^2$ in plan, with 60 mm of thickness. The fresh concrete was poured directly from
86 the mixing-truck by using a U-shape channel at the centre of the mould from a height of 60 cm. The influence of
87 fibre dispersion and orientation within the panel on the post-cracking behaviour was assessed by means of splitting
88 (Brazilian type) and direct tensile tests. Twenty-three specimens were extracted from each panel along the concrete
89 flow directions, according to the scheme represented in Fig. 1. In this figure the pale dash lines with arrows
90 represent the supposed concrete flow directions. When the drilling operations were performed, the panels were
91 already in their harden-mature phase. The hatched cores were used for executing splitting tensile tests, while the rest
92 were used for uniaxial tensile tests. In the splitting tensile test, to localize the specimen's fracture, two notches with
93 a 5 mm depth were executed on cores' opposite sides. The influence of the crack orientation towards the concrete
94 flow was assessed in two distinct directions. By assuming θ as the angle between the notched plane and the direction
95 of the concrete flow, the notch plane is designated parallel for $\theta = 0^\circ$ or perpendicular for $\theta = 90^\circ$. Since the core
96 scheme was maintained for both panels, for each core location there are two cores with perpendicular notch
97 direction. This will enable to evaluate the influence of fibre orientation at a certain distance from the casting position
98 on the stress-crack width (σ - w) relationship. For instance, θ of A1 specimen is 90° and 0° in panels A and B,
99 respectively.

100 The remaining cores extracted from the cast panels were sawn out from cylinders of 150 mm diameter and 60 mm
101 thickness according to the schematic representation shown in Fig. 2. Twenty two prismatic specimens with
102 dimensions of $110 \times 102 \times 60 \text{ mm}^3$ were produced for the uniaxial tensile test program. Following the same notching
103 procedure for the splitting test specimens, the prismatic specimens were notched according to parallel ($\theta = 0^\circ$) and
104 perpendicular ($\theta = 90^\circ$) directions to the expected concrete flow. The notch was executed in the four lateral faces of
105 the specimen, at its mid-height, with a thickness of 2 mm and a depth of 5 mm. Special care was given to this
106 operation to produce a notch with precise and uniform dimensions, and also to ensure the notch plane becomes
107 perpendicular to the direction of the applied stress.

108 **2.3 Test setup**

109 *2.3.1 Splitting tensile test*

110 To determine the σ - w relationship representative of the SFRSCC panel, splitting tensile tests based on the ASTM C-
111 496 [18] were executed. The tests were carried out in displacement-control using an universal testing rig with a
112 bearing capacity of 150 kN. The tests were performed with a relatively low displacement rate of 0.001 mm/s
113 enabling to obtain a stable response once the crack process is initiated. This low displacement rate was kept constant
114 throughout the test execution. An external displacement transducer positioned on the actuator that measured the
115 vertical deformation of the specimen was used to control the test.

116 Each specimen was positioned between two rigid supports and subjected to a diametral compressive line load
117 applied along the thickness of the specimen. It is assumed that this applied load induces a constant tensile stress in
118 the central part of the notched plane; therefore the results are expected to be close to the uniaxial tensile test results
119 [19]. The test setup is depicted in Fig. 3. In each specimen five linear variable differential transducers (LVDTs) were
120 applied according to the configuration schematically represented in Fig. 3a and 3b to record crack opening along the
121 notched plane. The support aluminium plates of each LVDT guarantee the register of the opening of the two
122 opposed faces of the notch, Fig. 3c. To assess if unsymmetric crack opening occurs, due to fibre segregation during
123 the casting procedure, two LVDTs were located at the specimen's bottom surface, while the others were fixed on the
124 top surface of the specimen.

125 *2.3.2 Uniaxial tensile test*

126 After sawing and notching operations, each specimen was carefully cleaned with pressurized air and acetone. Then,
127 two loading steel plates were glued with epoxy to the top and bottom surfaces of the specimen and subjected to a
128 uniform pressure during three days enabling the perfect alignment of the loading plates. Sikadur[®]-30 Normal epoxy
129 adhesive was used for this purpose.

130 A high stiff universal testing rig with a bearing capacity of 1000kN was used to execute the uniaxial tensile tests,
131 Fig. 4a. This test was carried out in close-loop displacement control by averaging the signals of four displacement
132 transducers installed on the two opposite faces of the specimen (top and bottom of the panel), Fig. 4b. Distinct
133 displacement rates were applied during the test according to the following procedure: 0.005 mm/min up to a
134 displacement of 0.05 mm, 0.02 mm/min up to a displacement of 0.1 mm, 0.08 mm/min up to a displacement of 0.5

135 mm/min and finally, 0.1 mm/min until the completion of the test. The adopted displacement rates comply with the
136 recommendations of RILEM TDF-162 [20].

137 2.3.3 Assessment of fibre distribution and orientation

138 To find out correlations between fibre distribution parameters and mechanical properties, such as, residual stresses
139 and absorption energy, it is quite important to determine fibre dispersion and fibre orientation parameters. There are
140 several methods for assessing the fibre distribution and orientation in fibre reinforced composites, namely:
141 tomography (CT-scan) [21], image analysis [22], x-ray method [17], electrical resistivity [17], ultrasound and
142 quantitative acoustic emission technique [23], and magnetic approach [24]. Among these methods, image analysis
143 technique was chosen due to its simplicity and relatively low cost of the necessary equipment.

144 The adopted procedure for fibre detection comprised four main steps. Firstly, the fracture surface of the specimen
145 was grinded. To enhance the reflective properties of the steel fibres, the surface was polished and cleaned with
146 acetone. Secondly, a colored image of this surface was taken using a high resolution digital photograph camera.
147 Afterwards, the obtained image was processed using ImageJ [25] software to recognize steel fibres. These steps are
148 depicted in Fig. 5. This method was also adopted by other researchers [13, 26, 27]. After analyzing the images, the
149 acquired data was processed, and the total number of fibres intersecting the plane (N_T^f), number of effective fibres
150 (N_{eff}^f), orientation of each fibre (θ), and segregation factor (ξ_{seg}) were obtained. Each parameter will be defined
151 subsequently.

152 The number of fibres per unit area, N^f , is the ratio between the total number of fibres counted in an image, N_T^f ,
153 and the total area of the image, A :

$$154 \quad N^f = N_T^f / A \quad (1)$$

155 The effective fibres, N_{eff}^f , per unit area are those that had the hooked end deformed, and those that have fractured.

156 The number of effective fibres was determined by visual inspection of the fracture surfaces.

157 The assessment of the fibre orientation degree at a certain plane can be given by a fibre orientation factor, η_θ , Eq.(2).

158 Based on an image analysis procedure of cut planes, the ellipses' axis of an intersecting fibre can be easily
159 determined. Therefore, in this method, the orientation factor η_θ can be regarded as an average orientation towards a
160 certain plane surface.

161
$$\eta_{\theta} = \frac{1}{N_T^f} \cdot \sum_{i=1}^{N_T^f} \cos \theta_i \quad (2)$$

162 In Eq. (2) N_T^f is the total number of fibres that can be determined by counting all the visible ellipses and circles at
 163 the cross section, θ is the out-plane angle that is defined as the angle between the fibre's longitudinal axis and a
 164 vector orthogonal to the plane.

165 The last analysed parameter was the fibre segregation along the gravity direction, determined from:

166
$$\xi_{seg} = \frac{1}{h \cdot N_T^f} \cdot \sum_{i=1}^{N_T^f} \bar{y} \quad (3)$$

167 where \bar{y} is the coordinate in the Y axis of the fibre's gravity centre, and h is the height (or depth) of the analysed
 168 cross-section. To calculate the location of the steel fibres gravity centre, an average value of the coordinates in the Y
 169 axis of entire fibres should be determined in the analysed cross-section. The ξ_{seg} parameter can assume values
 170 between 0 (segregation at the top of the cross-section) and 1 (segregation at the bottom of the cross-section). In a
 171 SFRC with ideal fibre distribution, ξ_{seg} is 0.5.

172

173

174 3. ANALYSIS OF RESULTS AND DISCUSSION

175 Table 2 includes the residual stresses and toughness parameters for different average crack widths. In this table, σ_{peak}
 176 is the stress at peak load that represents the maximum tensile stress; $\sigma_{0.3}$, σ_1 and σ_2 are the residual stresses at a crack
 177 opening width of 0.3, 1 and 2 mm, respectively; G_{F1} and G_{F2} are the dissipated energy up to a crack width of,
 178 respectively, 1 and 2 mm. Additionally, the coefficient of variation, CoV, and the characteristic values for a
 179 confidence interval of 95%, $k_{95\%}$, are also included. From the results it is noticed that the influence of the notch
 180 orientation towards the concrete's flow on the post-peak behaviour of the material is quite high. The series with a
 181 notch inclination of $\theta = 0^\circ$ shows higher residual tensile stresses and also larger dissipated energy than the specimens
 182 with $\theta = 90^\circ$. This variation in the post-cracking parameters could be ascribed to a preferential orientation of the
 183 fibres at the fracture surface. As it will be discussed in more detail further ahead, during the casting stage, fibres
 184 have the tendency to be aligned perpendicular to the direction of concrete flow, maybe due to a uniform radial

185 velocity profile as also observed by [14, 17]. Therefore, for the specimens with the notched plane parallel to the flow
186 direction, more fibres are almost perpendicular to the crack plane and, consequently, a higher number of fibres
187 intersect more effectively the fracture surface. Previous research on the fibre pullout behaviour has revealed that
188 fibre reinforcement effectiveness is almost the same for a fibre orientation towards the normal to the crack plane
189 lower than 30 degrees [28].

190

191 **3.1 Splitting tensile test**

192 Fig. 6 depicts the nominal stress – crack opening mouth displacement relationship, $\sigma - w$, for specimens extracted
193 from distinct panels' locations. The envelope and the correspondent average curves are presented in this figure. The
194 crack width was determined by averaging the recorded displacements of the 5 LVDTs installed on the faces of each
195 specimen, see Fig. 3. The nominal tensile stress at the centre of the specimen was obtained from the following
196 equation [29]:

$$197 \quad \sigma = \sigma_{sp} = \frac{2F}{\pi DL} \quad (4)$$

198 where F is the applied line load, D is the diameter of the cylinder (150 mm) and L is the thickness of the net area in
199 the notched plane (50 mm). Although the applicability of Eq. (4) is arguable in the softening phase of SFRSCC,
200 since it is based on the theory of elasticity, it will be used to estimate the tensile stress at the cracked surface, as
201 adopted by other researchers [19, 26, 30].

202 The $\sigma - w$ responses are almost linear up to the stress at crack initiation. Up to this stress level, the displacements
203 recorded by the LVDTs represent the transversal elastic deformation of the SFRSCC volume between the supports
204 of the LVDTs (Fig. 3c). Therefore, the deformability during this first phase should have been removed from the $\sigma -$
205 w response, but due to its negligible value this was not executed. After crack initiation, the $\sigma - w$ response is
206 nonlinear up to peak load. Once the peak load was attained, the load has smoothly decreased being visible a
207 softening response. Note that, for the specimens with the notch perpendicular to the flow direction ($\theta = 90^\circ$), the
208 peak stress was equal to the stress at crack initiation.

209 Generally, the $\sigma - w$ responses exhibited a relatively high scatter. In SFRSCC, this type of scatter is generally high,
210 even in specimens from the same casting and with the same testing conditions, due to the high dependence of the
211 post-cracking behaviour on the fibre distribution and orientation. Since the specimens were extracted from distinct

212 slab locations, at different distances from the casting point, a high scatter was expected. In fact, the viscous nature of
213 SFRSCC affects the distribution of the concrete constituents along the flow process.

214 Fig. 7 shows the $\sigma - w$ relationships at the two sides of the specimens, representative of the top and bottom surfaces
215 of the panels. Additionally the average curve is also included. The crack width was determined by averaging the
216 LVDTs readouts installed on each surface. As it is shown from the results, the LVDTs on the bottom surface
217 registered a lower value of the crack opening than the ones at the top surface for the same load level. This means
218 that the crack opened asymmetrically, which is justified by the fibre tendency to segregate along the depth of the
219 element [31]. The effect of fibre segregation was slightly higher in the $\theta = 90^\circ$ series. This aspect will be
220 corroborated and discussed in a subsequent section with the determination of a fibre segregation factor.

221

222 **3.2 Uniaxial tensile test**

223 Fig. 8 depicts the average and envelope stress-crack width ($\sigma-w$) curves regarding to each series. For both series ($\theta =$
224 0° and 90°), the $\sigma-w$ curve is almost linear up to the load at crack initiation. The concrete tensile strength was
225 approximately 2.7 MPa. Once the tensile strength is attained, the stress suddenly decreases up to a crack width about
226 0.07 mm. Beyond this crack width, $\theta = 0^\circ$ and 90° series behave in a completely distinct way. A semi-hardening and
227 a plateau responses are observed for the $\theta = 0^\circ$ and 90° series, respectively. Regarding the $\theta = 0^\circ$ series, Cunha *et*
228 *al.*[28] have analyzed the micromechanical behaviour of hooked end fibres by performing fibre pull-out tests, and
229 have verified that after a fibre sliding of nearby 0.1 mm, the fibre reinforcement mechanism is mainly governed by
230 the hook plastification during the fibre pull-out process. Therefore, in this series, fibres start to be pulled-out slowly
231 being observed a semi-hardening response. Afterwards, in $\theta = 0^\circ$ specimens, up to the crack width of about 0.6 mm,
232 a plateau response is observed, which is then followed by a smooth drop in the residual stress. From experimental
233 and analytical analysis, it was verified [32-33] that the average orientation angle value of the active fibres bridging a
234 leading crack is about 35° . According to fibre pull-out tests performed by Cunha *et al.* [34], in the case of the
235 inclination angle of 30° with the load direction, fibre rupture is the most predominant failure mode between the slip
236 range of 0.6-1.0 mm. In fact, during the uniaxial tensile test execution, after peak load is attained the sound of the
237 fibre rupturing was clearly noticeable that caused a rapid drop in the value of the load. This was confirmed after
238 analysing the fracture surface by visual inspection.

239 In the case of $\theta = 90^\circ$, some specimens shown a pseudo-hardening behaviour, especially those located nearby the
240 centre of the panel. After this pseudo-hardening behaviour, it is observed a small plateau followed by a reduction of
241 the residual stress beyond a crack width of about 0.9 mm, which corresponded to the rupture of the fibres.

242 The pre-peak branch shows very low scattering, while in the post-cracking phase the scatter of the response was
243 considerably higher. In the elastic phase the contribution of fibres is rather negligible. After crack initiation, the role
244 of the fibres becomes more important in bridging the stresses across the crack surfaces. This process depends
245 significantly on how fibres are distributed and oriented through the matrix, which means the scattering observed in
246 the post-cracking phase is highly influenced by the variation of the fibre dispersion and orientation amongst
247 different specimens. Hence, for the latter series it is more logical to categorize the σ - w relationships based on
248 distinct fibre orientation factor and distribution, which will be discussed in the next section.

249

250 3.3 Comparison of the results

251 Fig. 9 shows the relationship between the ratio of the splitting tensile post-cracking parameters, σ_{SPLT} , $G_{F\ SPLT}$ and the
252 uniaxial tensile post-cracking parameters, σ_{UTT} , $G_{F\ UTT}$ for the crack width corresponding to σ_{peak} that is known as
253 w_{peak} , and at crack width values of 0.3, 1.0 and 2.0 mm. In Fig. 9(a) for $\theta = 0^\circ$ series, w_{peak} does not represent the
254 same value for splitting tensile test (0.44 mm) and uniaxial tensile test (0.34 mm), therefore this interval is
255 represented as a hatched vertical strip. For the $\theta = 90^\circ$ series this problem is not crucial since σ_{peak} coincides with the
256 stress at crack initiation, which happened for a negligible crack opening (w_{peak}). The data plotted in Fig. 9(a) clearly
257 shows that σ_{SPLT} is larger than σ_{UTT} for almost all w ($CMOD$) values considered except at $w = w_{peak}$ for the $\theta=90^\circ$
258 series. Therefore, splitting tensile test overestimates the tensile residual strength. The average tensile stress at peak
259 load for the splitting and uniaxial tensile test was 4.39 and 3.30 MPa for $\theta = 0^\circ$ specimens, and 2.47 and 2.72 MPa
260 for $\theta=90^\circ$ series. With the increase of the crack opening, the $\sigma_{SPLT} / \sigma_{UTT}$ ratio became higher, since in the softening
261 phase fibres started being mobilized as they bridge the stresses across the crack surfaces.

262 Fig. 9(b) depicts the relationship between the energy absorbed during the fracture process in both test setups, up to a
263 crack width of 0.3, 1 and 2 mm. Both series presented a similar tendency, an increase of G_f with the crack width was
264 observed. On the other hand, in the average term, for 0.3 mm crack width, the
265 $G_{F\ SPLT} / G_{F\ UTT}$ ratio is 1.33 and 1.94 for $\theta = 0^\circ$ and $\theta = 90^\circ$ series, respectively. This ratio has increased up to 1.62
266 and 2.05 for 2 mm crack width, respectively, for $\theta = 0^\circ$ and $\theta = 90^\circ$ series.

267 3.4 Fibre distribution and orientation

268 Table 3 includes the fibre distribution parameters obtained by image analysis on the plane surface (see Fig. 10) of
269 the specimens subjected to uniaxial tension test. Within each panel, by assuming the casting point as origin,
270 specimens with the same distance from casting origin are presented in the same row. For each studied distance, the
271 number of fibres was assessed in two perpendicular planes ($\theta = 0^\circ$ and 90° , Fig. 1). From the analysed results, N^f
272 and N_{eff}^f were significantly higher at the specimens with $\theta = 0^\circ$, approximately 80% and 254 %, respectively, when
273 comparing to specimens with $\theta = 90^\circ$. This high variation of the fibre distribution in two perpendicular directions
274 could be ascribed to a preferential fibre alignment influenced by the concrete's flowability. Moreover, the
275 probability that a random section plane crossing a single fibre is a function of the fibre's length (L), diameter (D),
276 and also the angle that the it makes with the section plane (fibre orientation factor) [35]. Since all the fibres have the
277 same aspect ratio, the value of D and L are constant, therefore the probability function depends on the fibre
278 orientation factor. On the other hand, the higher orientation factor leads to a higher probability of a single fibre
279 intersecting a section plane. Concerning the fibre segregation factor, the obtained average values of ξ_{seg} for the
280 studied cross sectional planes were slightly higher than 0.5, approximately 7.6 to 14.6%. The obtained values are
281 coherent with the $\sigma - w$ curves depicted in Fig. 7, since it justifies why the value of the crack opening determined in
282 the bottom surface of the cores in $\theta = 90^\circ$ specimens is lower than the other series. Thus, for the studied self-
283 compacting concrete composition, slightly fibre segregation towards the bottom of the specimen due to the gravity
284 action was observed. In terms of the fibre orientation factor, η_θ , specimens from series $\theta = 0^\circ$ had higher values than
285 the $\theta = 90^\circ$ series, which means that the fibres are more aligned perpendicular to the fracture plane in the $\theta = 0^\circ$
286 series.

287 Fig. 11 depicts orientation profiles obtained for the average orientation factor of each series separately. In this
288 figure, the distribution of the orientation angle through the cut plane was studied for each specimen separately and
289 the experimental results were compared to Gaussian distribution. According to this study, the distribution of the
290 orientation angle follows closely a Gaussian distribution. Laranjeira *et al.* [36] had already obtained similar
291 conclusion. Based on this method, an Excel spreadsheet was developed in order to determine the probability density
292 distribution of fibre orientation. As it is expected, $\theta = 0^\circ$ specimens show a distribution shifted to the left side, which
293 means fibres have a tendency to be oriented more perpendicular to the cut plane (crack plane). On the other hand,

294 the $\theta = 90^\circ$ distribution is slightly transferred to the right side and more fibres tend to be aligned parallel to the cut
295 plane (crack plane). Regarding to the comparison with theoretical orientation values for a two-dimensional
296 distribution, $2/\pi$, [37] and a three-dimensional isotropic uniform random fibre distribution, 0.5, [38] $\theta = 0^\circ$
297 specimens had a very different distribution profile, whereas orientation profile in $\theta = 90^\circ$ series has matched with 2D
298 fibre random distribution perfectly. Consequently, in the case of casting panels from the centre, for $\theta = 0^\circ$ series the
299 assumption of a 2D or 3D uniform random fibre distribution is far apart from the reality. In the present case, the
300 distribution is prominently influenced by the placing conditions and concrete flowability.

301 Based on the obtained results, since in the casting process of the panels, particularly from the centre, the wall effects
302 are negligible, the flow velocity is uniform and diffuses outwards radially from the casting point, see Fig. 12.
303 Therefore, fibres have a tendency to orient perpendicular to the concrete flow direction. As a consequence, in the $\theta =$
304 0° series the SFRSCC presented a semi-hardening response due to the high number of effective fibres with
305 favourable orientation, while in the $\theta = 90^\circ$ series, since fibres were rotated due to the concrete flow velocity, the
306 number of the effective fibres is reduced and lower residual strengths are observed.

307 Fig. 13 depicts the relationship between the fibre density measured at the notched fracture surfaces after performing
308 direct tensile test and the distance from the casting point. In this figure $N_{f_{\parallel}}$ and $N_{f_{\perp}}$ are, respectively, the fibre
309 density at a crack plane parallel and perpendicular to the concrete flow. As it is expected, due to the proper viscosity
310 of the concrete, a good homogeneity and dispersion of the fibres were achieved all over the panels, and a higher
311 fibre density was obtained in the fracture surfaces in the alignment of the concrete flow.

312 The $\sigma - w$ relationships previously obtained (see Fig. 8) have shown a high scatter due to the distinct fibre
313 distributions. In order to reduce the scatter of the results and also study the influence of η_θ and N_{eff}^f , the $\sigma - w$
314 relationships were separated in three different categories, see Fig.14. From this figure, it is concluded that the post-
315 cracking parameters depend not only in η_θ but also in N_{eff}^f . Fig. 15 clearly shows that by increasing the orientation
316 factor, the number of the effective fibres tends to rise exponentially.

317

318

319 **4. CONCLUSION**

320 In the present work, the influence of fibre distribution / orientation on the tensile performance of steel fibre
321 reinforced self-compacting concrete (SFRSCC) was characterized by performing splitting and uniaxial tensile tests
322 on cored specimens extracted from different panel locations.

323 Fibre distribution and orientation have a strong impact on the tensile behaviour of specimens drilled from the panels.

324 In the case of the series with crack plane parallel to the concrete flow direction ($\theta = 0^\circ$), specimens shown
325 significantly higher post-cracking parameters than the other studied case with a perpendicular crack plane to the
326 flow direction ($\theta = 90^\circ$). When a panel is cast from the centre, fibres have a tendency to line up perpendicularly to
327 the radial flow, mainly due to the uniform flow profile velocity that diffuses outwards radially from the centre of the
328 panel. Hence, the total number of the effective fibres intersecting the parallel crack plane ($\theta = 0^\circ$) was higher than
329 the one registered in the orthogonal crack plane ($\theta = 90^\circ$).

330 The probabilistic distribution of the orientation angle through a cut plane follows closely a Gaussian distribution. By
331 determining the probability density function of fibre orientation for each series separately, it is found that for $\theta=0^\circ$
332 specimens the assumption of 2D and 3D uniform random fibre distribution is completely far apart from the reality,
333 while $\theta = 90^\circ$ series follows a pattern very close to the theoretical 2D random fibre distribution.

334 Splitting tensile tests tend to overestimate the post-cracking parameters, but clearly capture all phases of post-
335 cracking response. Moreover, the splitting tests have presented a lower scattering of the results when compared to
336 the uniaxial tensile test. The load at crack initiation step was not influenced by fibres; both tests estimated similar
337 tensile strengths. The post-peak stresses and energy absorption parameters obtained from the splitting tensile tests,
338 especially, the energy absorption parameters have shown a reasonable correlation with the ones obtained from the
339 uniaxial tensile tests.

340

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FIGURES AND TABLES

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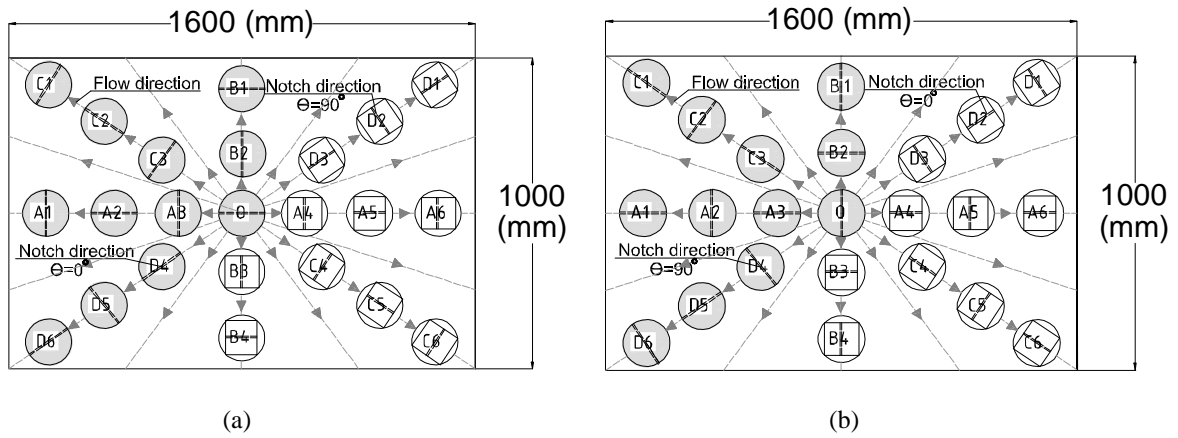
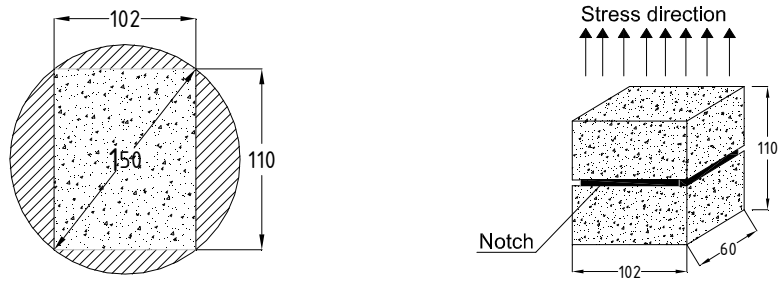


Fig. 1 - Core extracting plan: (a) panel A, (b) panel B.

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Fig. 2 - Schematic representation of the prismatic specimen production from an extracted core (dimensions are in

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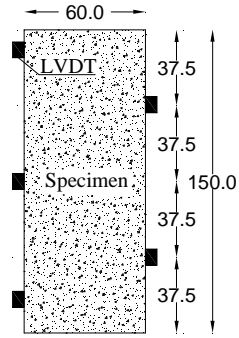
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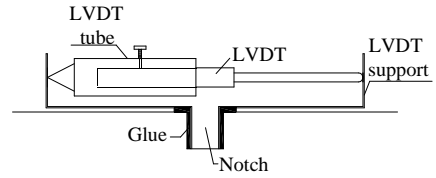
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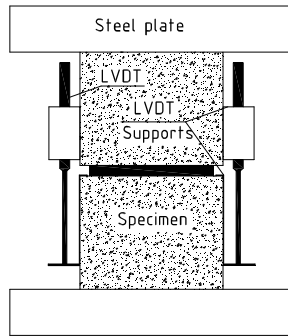
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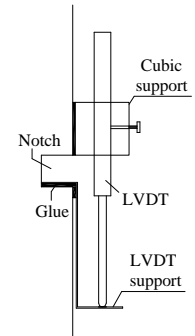
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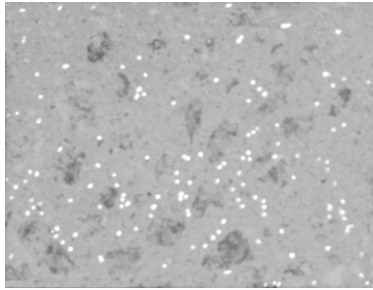
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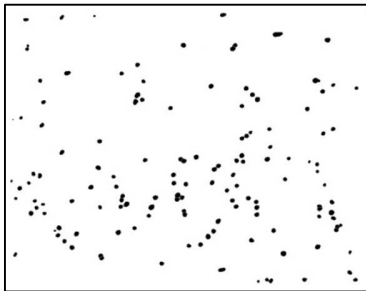
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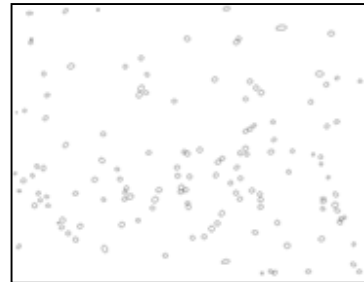
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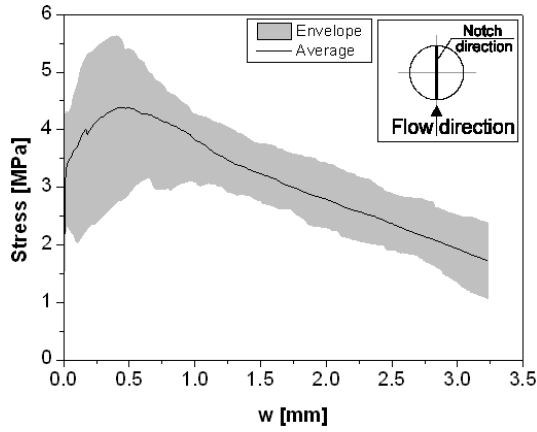


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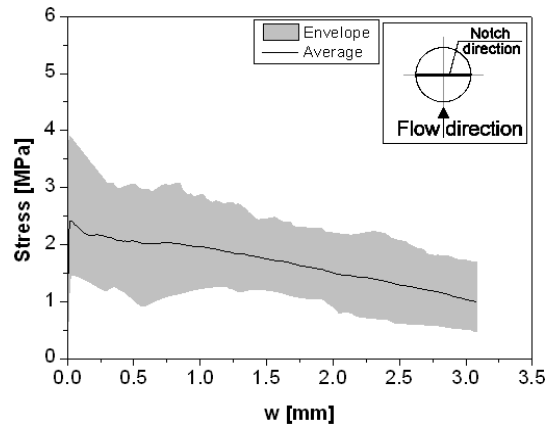


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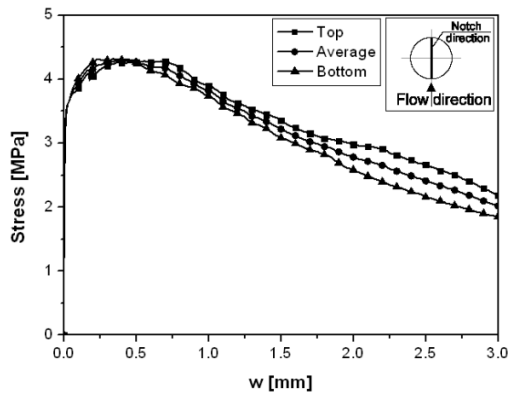
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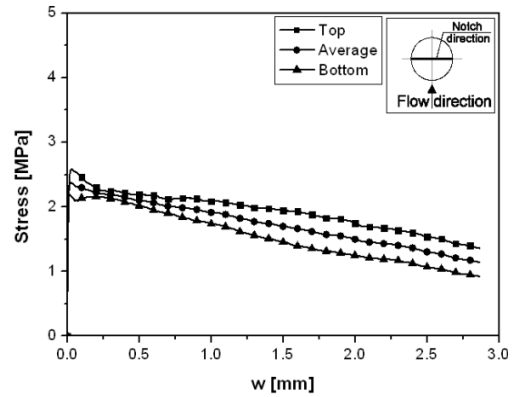
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Fig. 6 - Nominal tensile stress – crack opening width relationship, $\sigma - w$, obtained from splitting tensile test for: (a) $\theta = 0^\circ$ and (b) $\theta = 90^\circ$.

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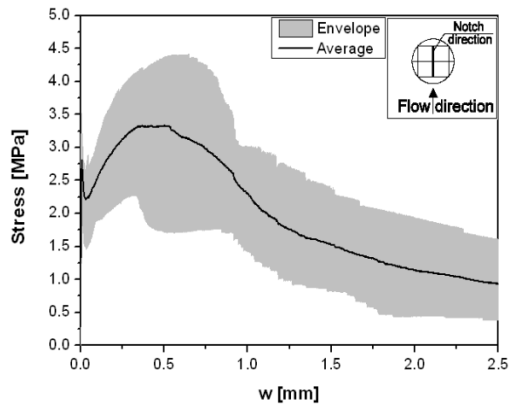
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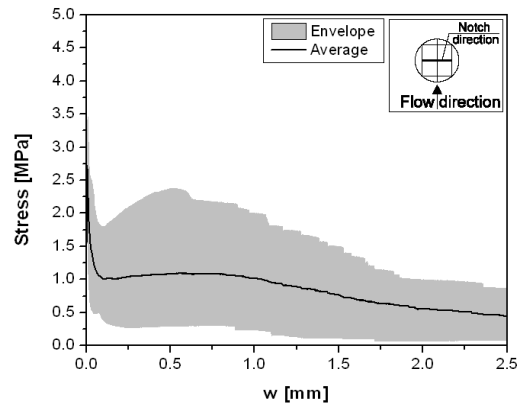
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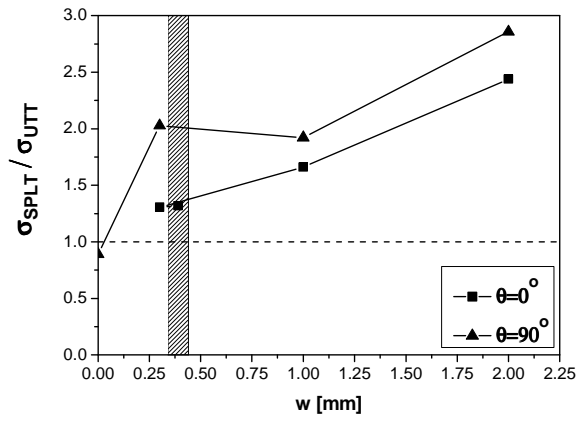
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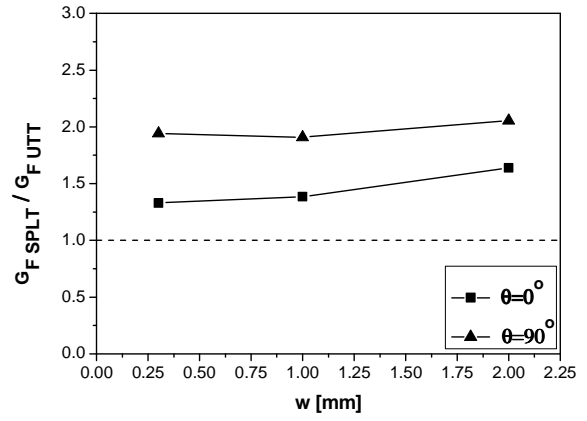
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Fig. 8 - Uniaxial tensile stress – crack width relationship, $\sigma - w$: (a) $\theta = 0^\circ$ and (b) $\theta = 90^\circ$.

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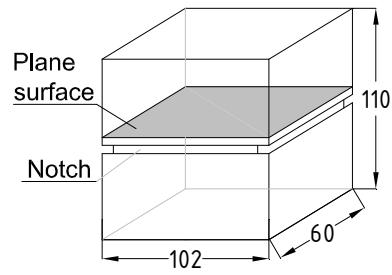
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Fig. 9 - Uniaxial tensile post-cracking parameters versus splitting tensile post cracking parameters: (a) Residual stress and (b) Fracture energy.

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675 Fig. 10 - Localization of the plane surface considered in the fibre distribution assessment (units in mm).

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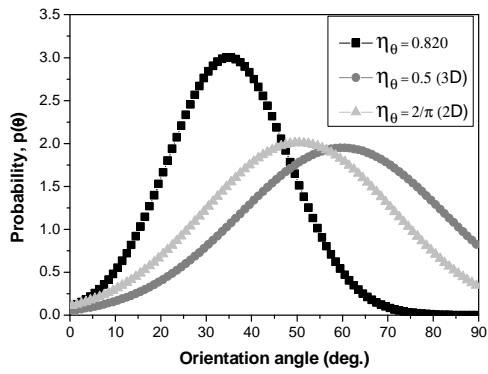
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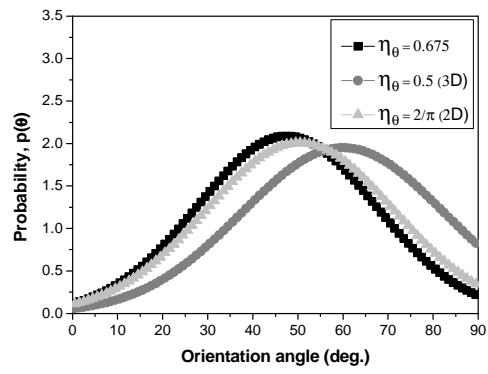
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Fig. 11 - Predicted orientation profile: (a) $\theta = 0^\circ$ and (b) $\theta = 90^\circ$.

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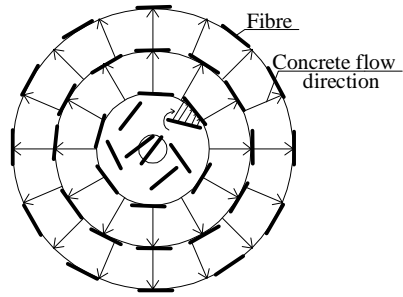
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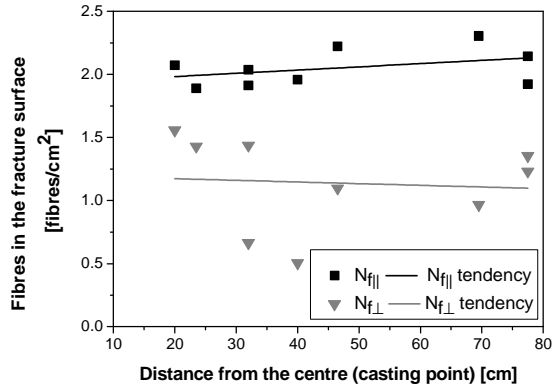
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Fig. 12 - Explanation for fibre alignment in flowing concrete of a panel casting from the centre.



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Fig. 13 - Relationship between the number of fibres in the fracture surfaces and the distance from the centre.

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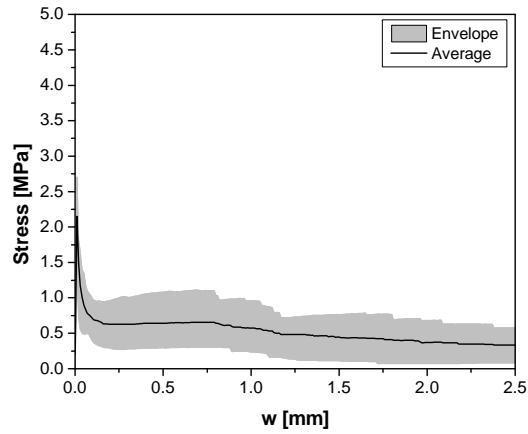
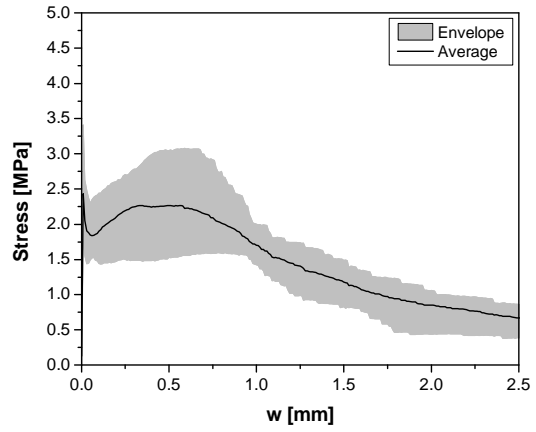
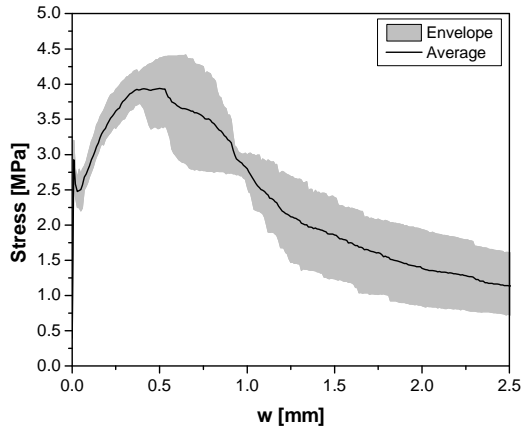
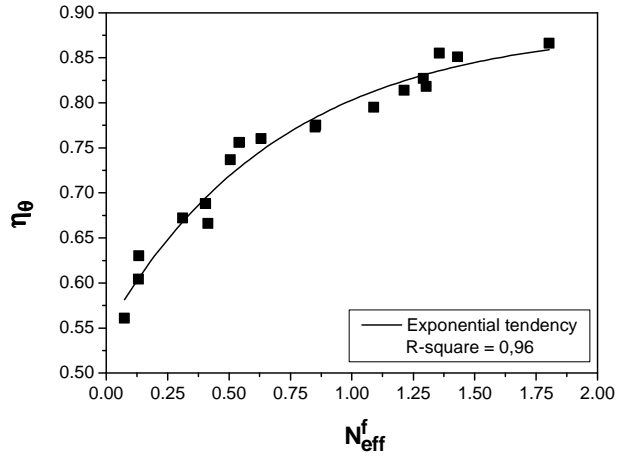


Fig. 14 - Categories of uniaxial tensile stress – crack width relationships, $\sigma - w$, : (a) $\eta_{\theta} \geq 0.80$ and $N_{eff}^f \geq 1.20$, (b)

$0.68 < \eta_{\theta} < 0.80$ and $0.41 < N_{eff}^f < 1.20$, (c) $\eta_{\theta} \leq 0.68$ and $N_{eff}^f \leq 0.41$.



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Fig. 15 - Orientation factor, η_θ , versus number of the effective fibres, N_{eff}^f .

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Table 1- Mix proportions of steel fibre reinforced self-compacting concrete per m³.

Cement [kg]	Water [kg]	W/C [-]	SP [kg]	Filler [kg]	Fine sand [kg]	Coarse sand [kg]	Coarse aggregate [kg]	Fibre [kg]
413	140	0.34	7.83	353	237	710	590	60

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Table 2 - Residual stress and toughness parameters obtained from splitting and direct tensile tests.

	Series	Parameter	σ_{peak} [MPa]	$\sigma_{0.3}$ [MPa]	σ_1 [MPa]	σ_2 [MPa]	G_{F1} [N/mm]	G_{F2} [N/mm]
Splitting tensile test	$\theta = 0^\circ$ $(\sigma_{\parallel})^*$	Average	4.39	4.23	3.82	2.79	4.07	7.32
		CoV(%)	25.6	29.7	24.3	30.2	27.2	25.2
		K _{95%}	3.52	3.16	2.09	1.95	3.36	6.08
	$\theta = 90^\circ$ $(\sigma_{\perp})^*$	Average	2.47	2.13	1.96	1.50	2.08	3.82
		CoV(%)	33.1	48.6	37.9	35.3	35.9	33.2
		K _{95%}	2.07	1.74	1.46	1.09	1.49	2.83
Uniaxial tensile test	$\theta = 0^\circ$ $(\sigma_{\parallel})^*$	Average	3.33	3.24	2.30	1.14	2.94	4.47
		CoV(%)	19.0	21.4	27.4	39.8	24.2	23.7
		K _{95%}	3.10	2.73	1.83	0.80	2.42	3.72
	$\theta = 90^\circ$ $(\sigma_{\perp})^*$	Average	2.72	1.05	1.02	0.56	1.09	1.86
		CoV(%)	19.1	64.5	65.4	57.1	59.6	59.9
		K _{95%}	2.34	0.51	0.48	0.30	0.57	0.96

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* \parallel and \perp - notch direction parallel ($\theta = 0^\circ$) and perpendicular ($\theta = 90^\circ$) to the concrete flow direction, respectively.

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Table 3- Fibre distribution parameters.

Specimen	Distance [cm]	$\theta = 0^\circ$				$\theta = 90^\circ$			
		N^f [fibres/cm ²]	N_{eff}^f [fibres/cm ²]	η_θ [-]	ξ_{seg} [-]	N^f [fibres/cm ²]	N_{eff}^f [fibres/cm ²]	η_θ [-]	ξ_{seg} [-]
B3	20.0	2.071	1.291	0.827	0.580	1.557	0.405	0.688	0.476
A4	23.5	1.889	1.356	0.855	0.518	1.430	0.506	0.737	0.510
C4	32.0	2.036	1.430	0.851	0.555	0.665	0.133	0.630	0.597
D3	32.0	1.913	0.853	0.775	0.491	1.436	0.415	0.666	0.586
B4	40.0	1.956	0.851	0.773	0.530	0.506	0.074	0.561	0.643
A5	46.5	2.220	1.212	0.814	0.479	1.097	0.311	0.672	0.725
A6	69.5	2.304	1.803	0.866	0.557	0.967	0.132	0.604	0.539
C6	77.5	2.142	1.303	0.818	0.600	1.232	0.541	0.756	0.485
D1	77.5	1.921	1.089	0.795	0.532	1.355	0.631	0.760	0.594
Average		2.050	1.24	0.820	0.538	1.138	0.35	0.675	0.573
CoV (%)		7.16	23.74	4.15	7.33	31.98	57.11	10.20	14.00

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